

THREE DIMENSIONAL MICROSTRUCTURAL CHARACTERIZATION OF CATHODE DEGRADATION IN SOFCs USING FOCUSED ION BEAM AND SEM

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THE DEPARTMENT *of*
MATERIALS SCIENCE AND ENGINEERING



Introduction to Solid oxide fuel cells

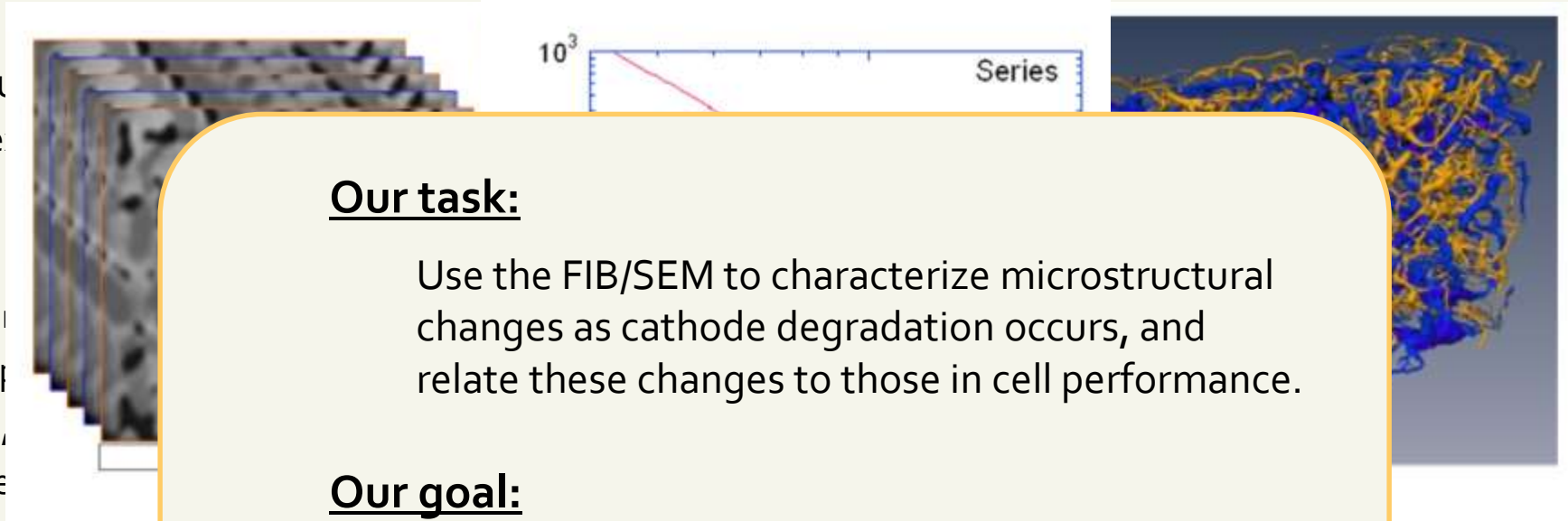
- Solid oxide fuel cells
 - Low cost, flexible

- Problems?
 - High operating temperatures
 - Cathode poisoning
 - H_2O , CO_2 , SO_2
 - Adverse effects on performance

- What is primary cause of degradation?

- **Microstructure!**

- Previous work:
 - Quantification in the FIB/SEM:
 - J. Wilson, S. Barnett, *Electrochem. Commun.*, **11**(5), 1052 (2009).
 - D. Gostovic, E. Wachsman, *et al.*, *J. Am. Ceram. Soc.*, **94**(2), 620 (2011).
 - Relationship to cell performance:
 - J. Smith, E. Wachsman, *et al.*, *Solid State Ionics*, **180**(1), 90 (2009).



Our task:

Use the FIB/SEM to characterize microstructural changes as cathode degradation occurs, and relate these changes to those in cell performance.

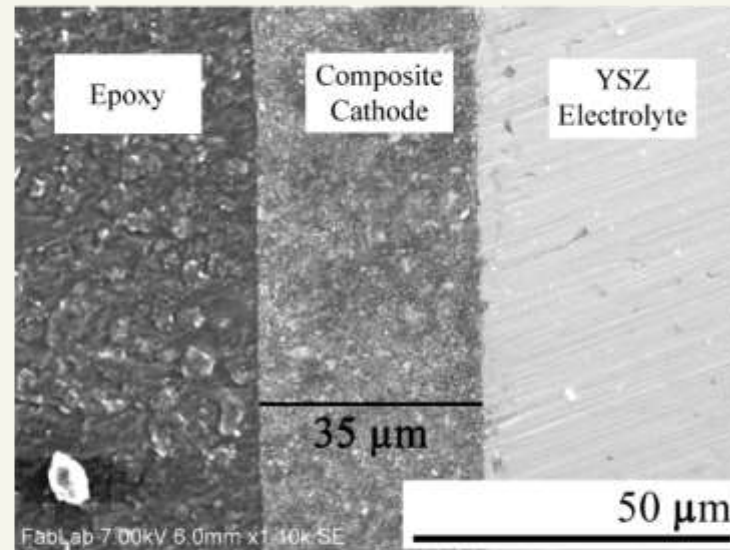
Our goal:

Better understanding of the fundamental mechanisms behind cathode degradation.

Experimental - Button cell testing



Button cell



Cross-section view

- Symmetric cathode cells
- 8-YSZ electrolyte
- 50 wt. % LSM/YSZ cathode paste
- Sintered at 1000°C for 1hr

- Three conditions compared:
 - Baseline ("*Unaged*")
 - Aged at 800°C for ~400 hr under ambient ("*Aged - dry*")
 - Aged at 800°C for ~400 hrs under 3% H₂O ("*Aged - H₂O*")

Data acquisition

- Our results (and conclusions) can only be so good as our inputs
 - We need good inputs! (GIGO)
- Important considerations:
 - Initial sample preparation (pre-FIB)
 - Sample preparation within the FIB/SEM
 - Slicing resolution (for fidelity of reconstruction)
 - Electron beam parameters - image noise and resolution vs. data acquisition time
 - What is it we need to accentuate?

Data acquisition – pre-FIB preparation

1) Vacuum epoxy impregnation:



2) Polishing (to 1200 grit):



3) Mounting for FIB/SEM:



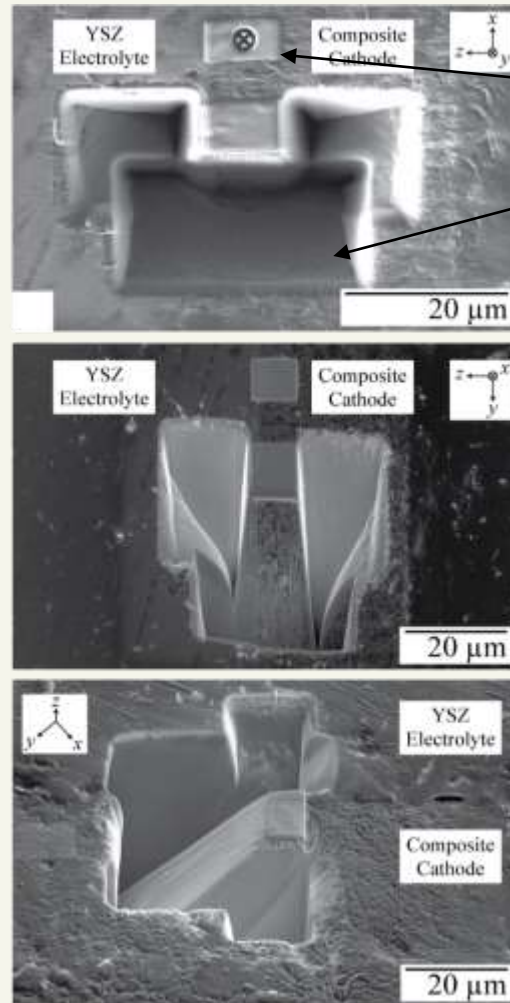
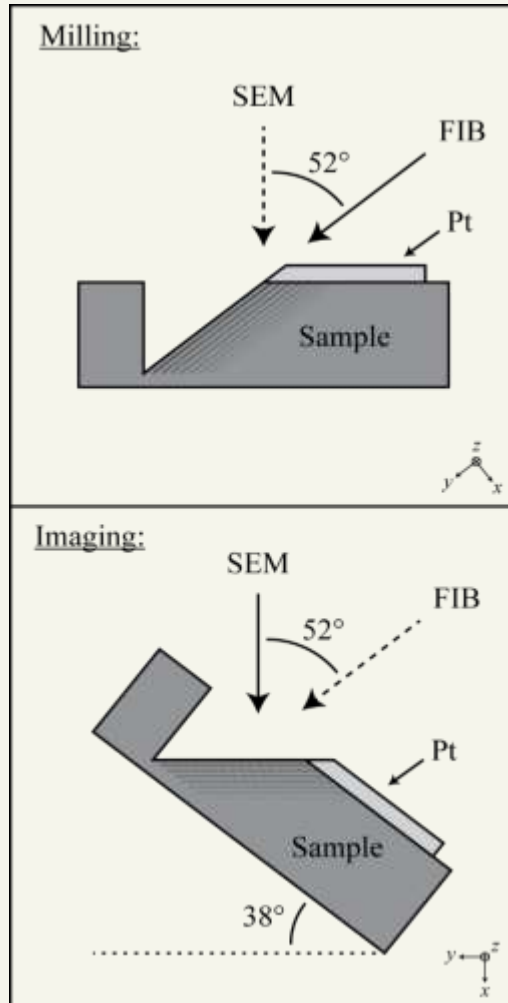
Equipment/Software

- FEI Helios 650
 - Part of the Center for Nanoscale Science and Technology (CNST) user facility at NIST
 - Multichem, iFast Developer Kit, etc.
- Auto Slice and View version 1.2
- Avizo Fire



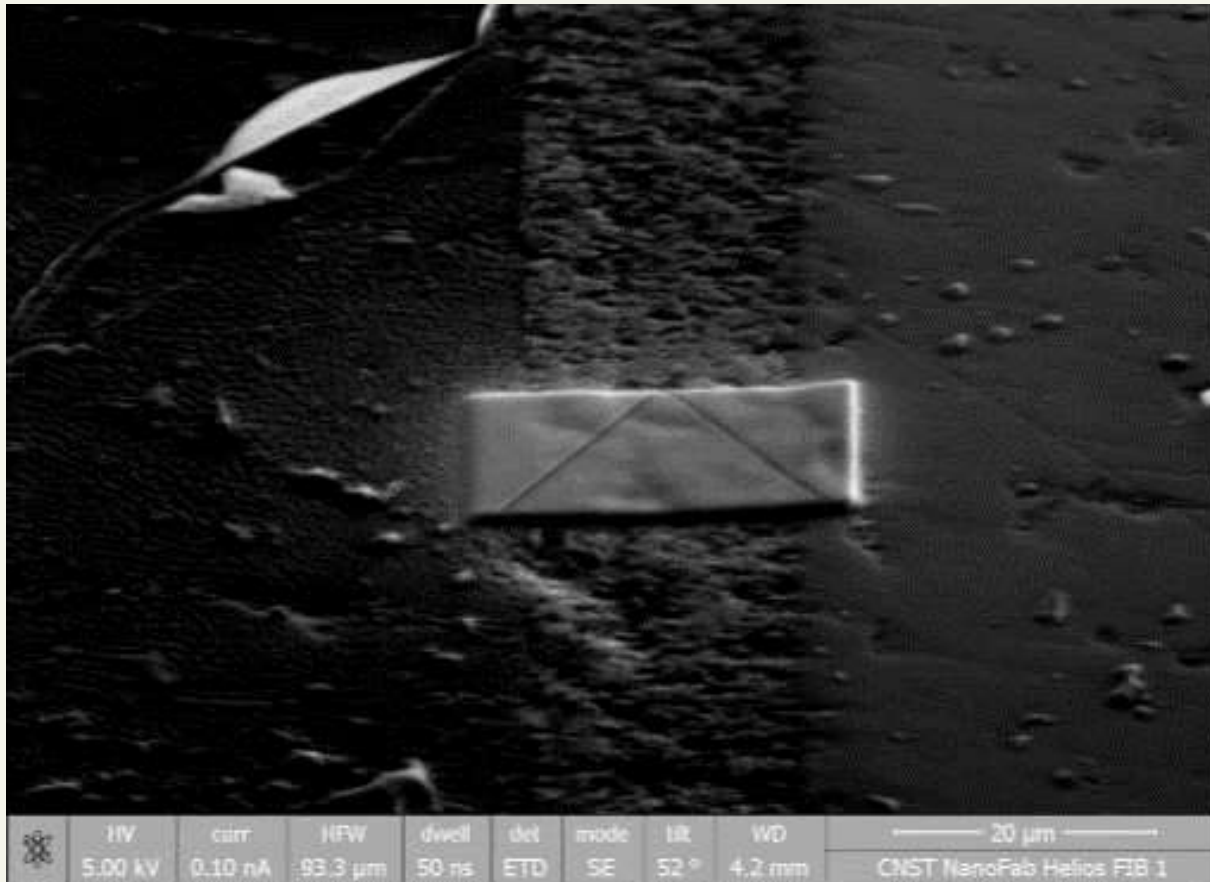
FEI Helios 650 at NIST (CNST)

Data acquisition – Initial procedures



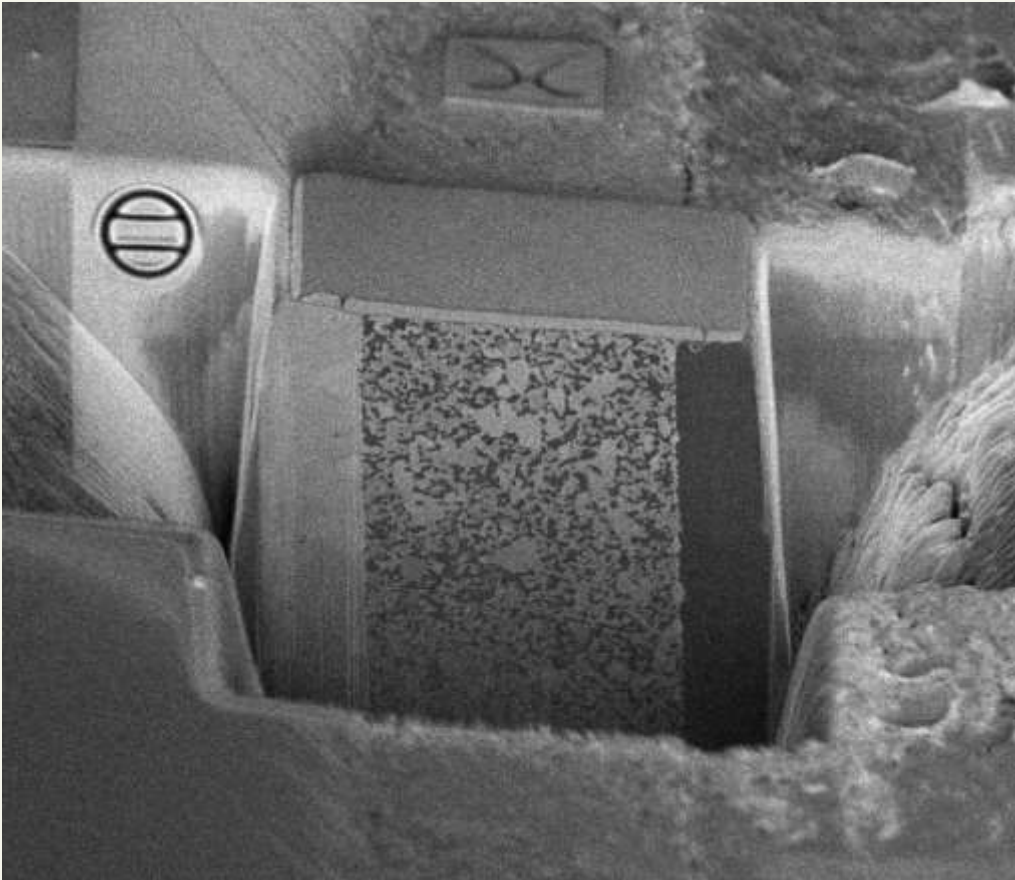
- Protective Pt and fiducial marks
- “C-trench” to prevent shadowing
- Mill at 0° stage tilt
- Image at 38° stage tilt
- Unique FIB/SEM geometry means:
 - Improved contrast between LSM and YSZ
 - Perpendicular imaging
 - Limited detector shadowing effect
 - **Increased acquisition time (significant!)**
 - 5 min 40 sec, per slice
 - 30 nm slice thickness, 20 μm slice width

Data acquisition – Optimized procedures



- Similar trenching, but in usual orientation
- iFast recipe developed to fully automate (takes about 1 hour to complete)
 - Pt deposition
 - Thickness tracking mills
 - C deposition (x2 for fiducial)
 - Fiducial mill
 - Bulk trenches (65 nA)
- Manually mill imaging fiducial and clean up face before beginning acquisition

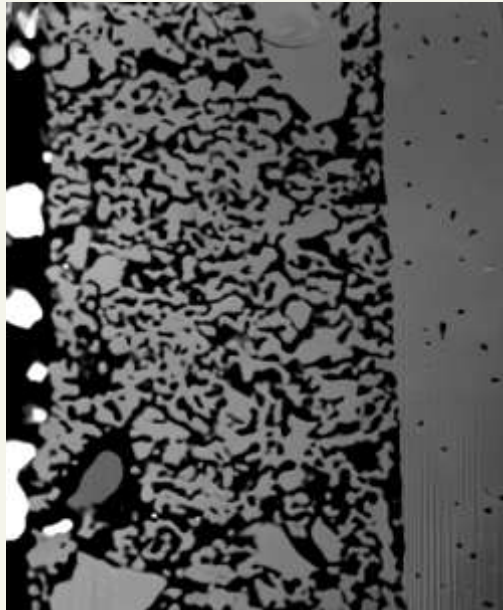
Data acquisition – Optimized procedures



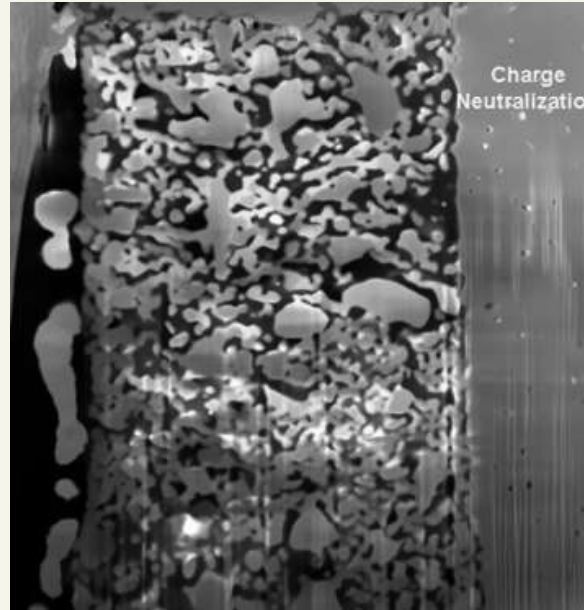
- Setup and ready to mill in about 1.5 hrs
 - Mostly automated
- Electron image fiducial for precise image placement
- Tracking marks for post-run measurement of slice thickness
- Some shadowing deeper into trench, but much faster acquisition
 - **2 min 30 sec, per slice**
 - 20 nm slice thickness, ~ 30 μm slice width
 - Overnight run acquires about 7 μm of depth

Experimental – Electron imaging challenges

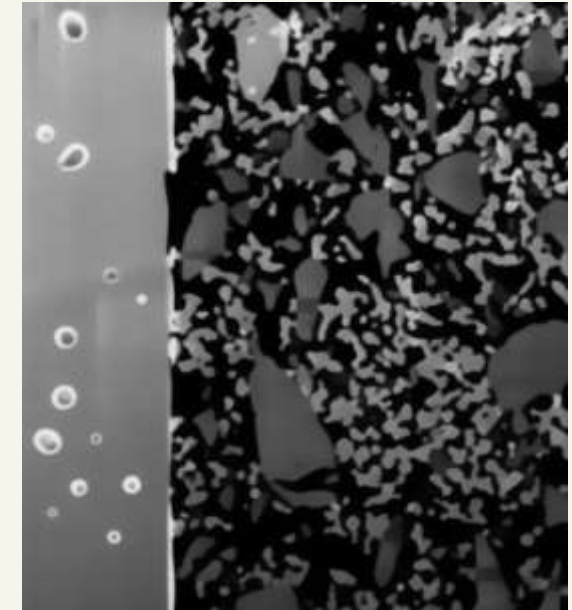
- Challenging system to image due to:
 - Poor RT conductivity
 - Similar electron yields from each phase
- Careful control of imaging parameters allows for charge contrast imaging
 - Not always completely reliable due to local effects



In-lens BSE

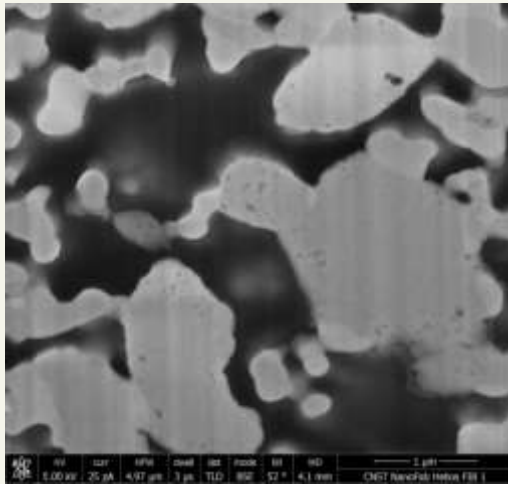


In-lens SE (Charge neutralization setting)



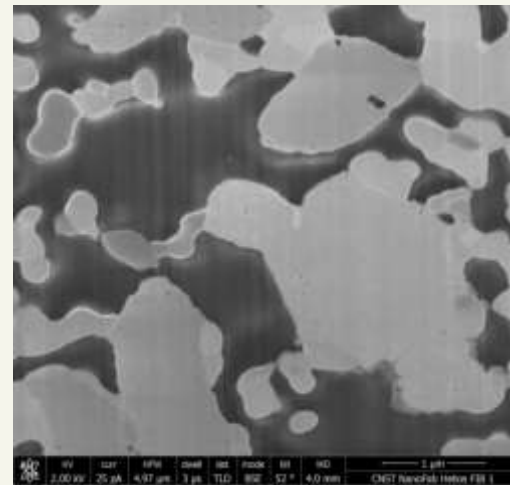
Long-dwell charging contrast (SE)

Experimental – Electron imaging (effect of V_{acc})

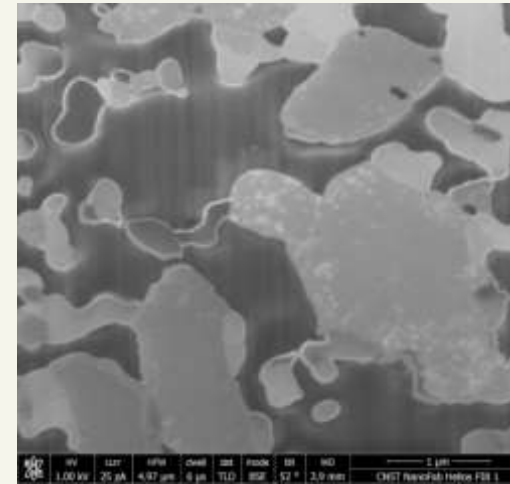


5kV – 25 pA

What contrast is really important?
Need to facilitate segmentation!



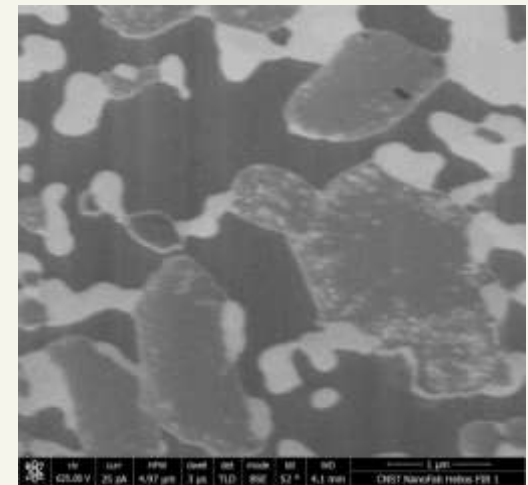
2kV – 25 pA



1kV – 25 pA

Image frame integration and longer dwell (6 μs) improve contrast between phases

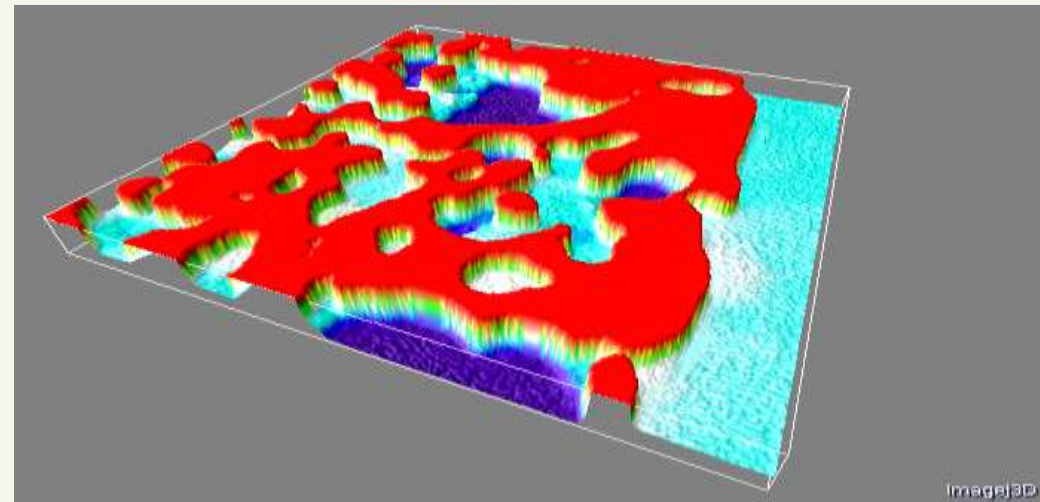
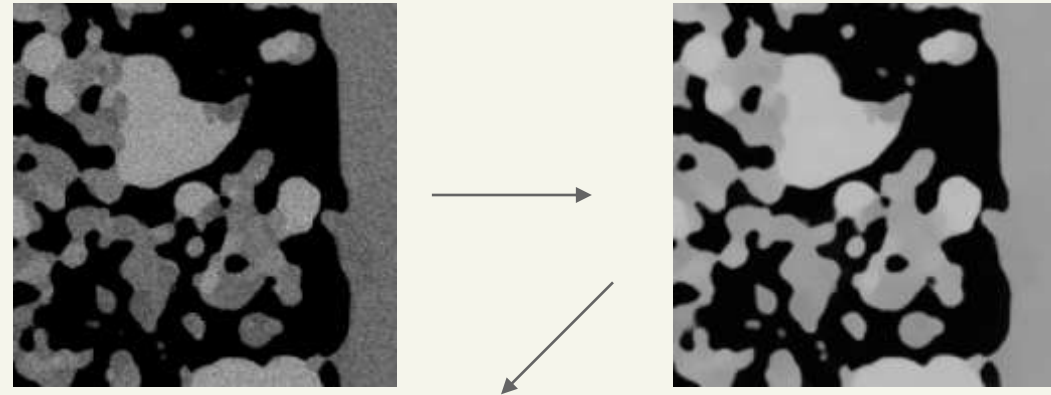
TLD
(through the lens detector)
in backscatter electron mode



625 V – 25 pA

Experimental – post processing of data

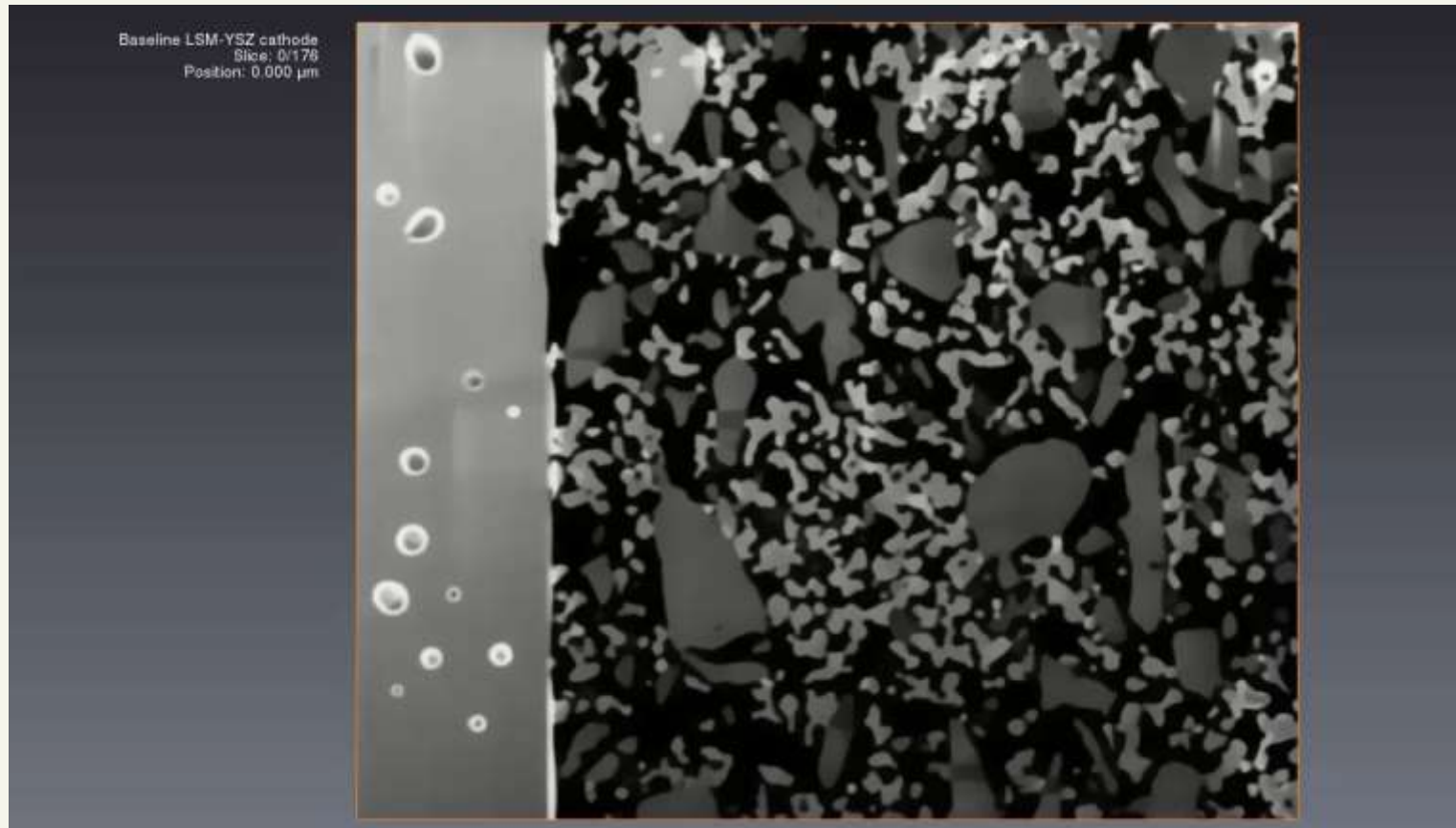
- Post-processing done with Avizo Fire:
 - Non-local means filtering of data¹:
 - Watershed segmentation algorithm²:



¹Based on A. Buades *et al.* in 2005 *IEEE Comput. Soc. Conf. Comput. Vis. Pattern Recognit.*, Vol. 2, p. 60. IEEE.

²L. Vincent and P. Soille, *IEEE Trans. Pattern Anal. Mach. Intell.*, 13(6), 583 (1991).

Results – *Unaged* SOFC



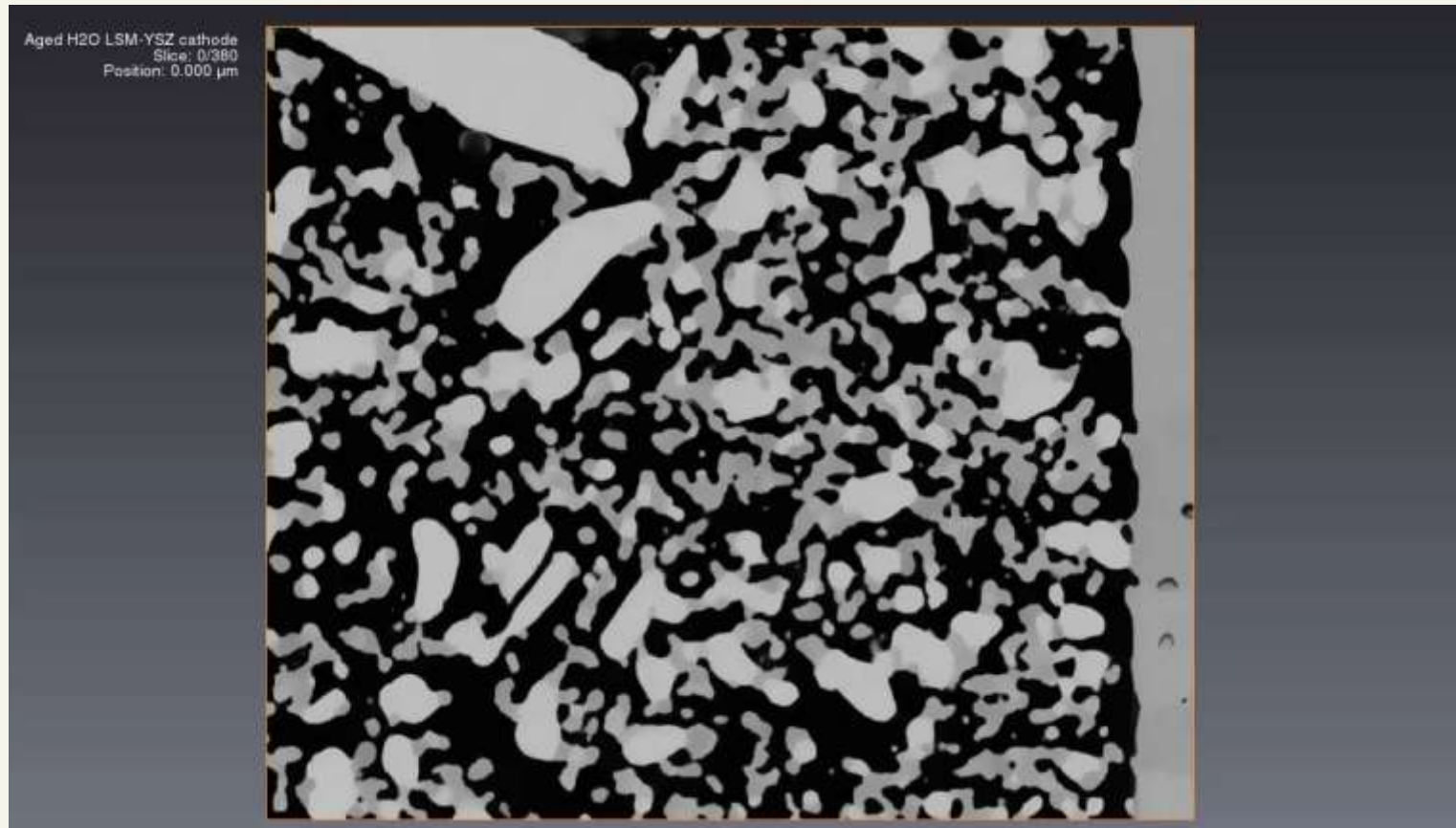
Voxel size: 12 x 12 x 30 nm, SE signal collected on FEI Helios 650, NIST
HFW: 9.5 μm ; Depth: 5.25 μm ; Total reconstructed cathode volume (excluding electrolyte): 276.84 μm^3

Results – *Aged, Dry SOFC*



Voxel size: 10 x 10 x 10 nm, EsB signal collected on Zeiss Crossbeam 540, courtesy of Carl Zeiss and Fibics, Inc.
HFW: 9.9 μm ; Depth: 8.12 μm ; Total reconstructed cathode volume (excluding electrolyte): 643.11 μm^3

Results – *Aged, H₂O* SOFC

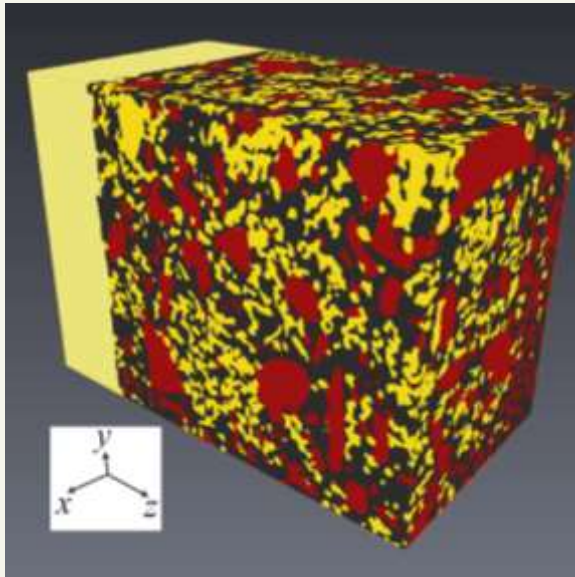


Voxel size: 10 x 10 x 10 nm, EsB signal collected on Zeiss Crossbeam 540, courtesy of Carl Zeiss and Fibics, Inc.
HFW: 13.96 μm ; Depth: 3.8 μm ; Total reconstructed cathode volume (excluding electrolyte): 627.35 μm^3

Results – Surface generation



Baseline

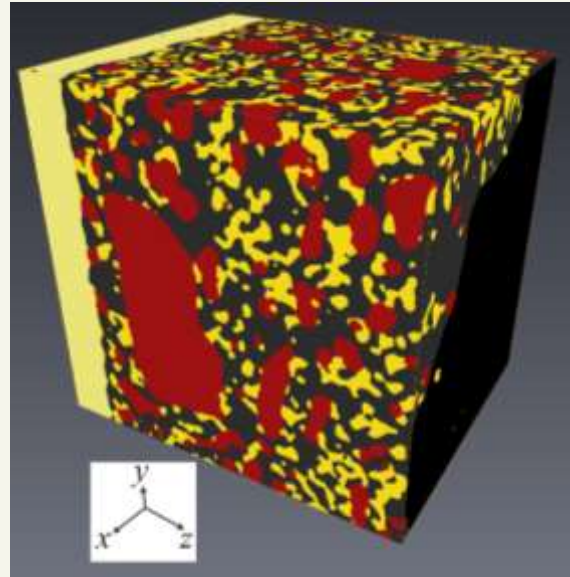


350.8 μm^3

Bounding box dimensions (μm):

5.250	7.031	9.504
X	Y	Z

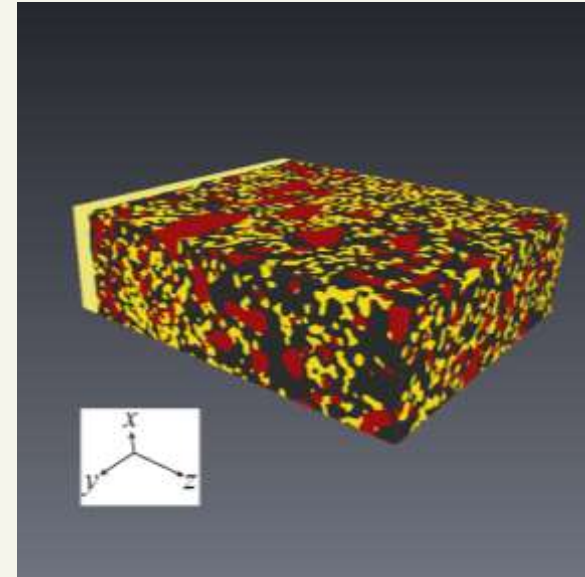
Aged-dry



761.1 μm^3

8.121	9.380	9.991
X	Y	Z

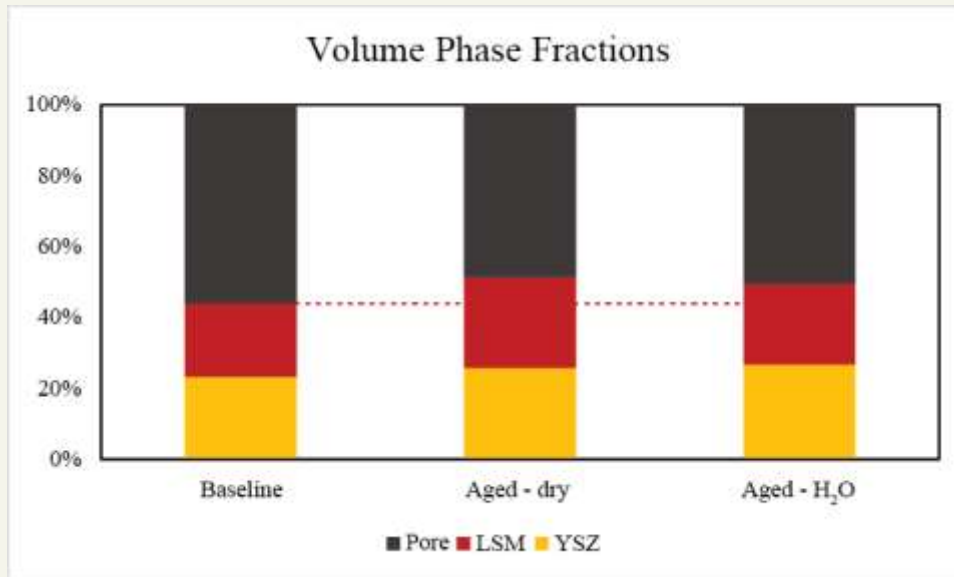
Aged-H₂O



667.2 μm^3

4.003	11.940	13.960
X	Y	Z

Results – Surface quantification

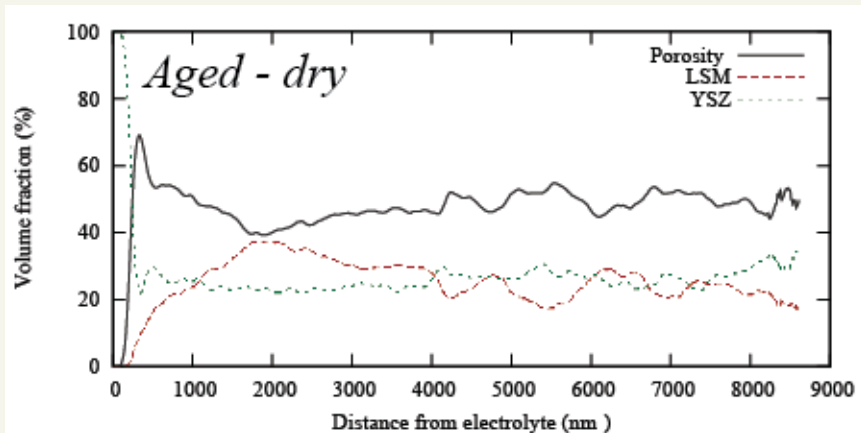
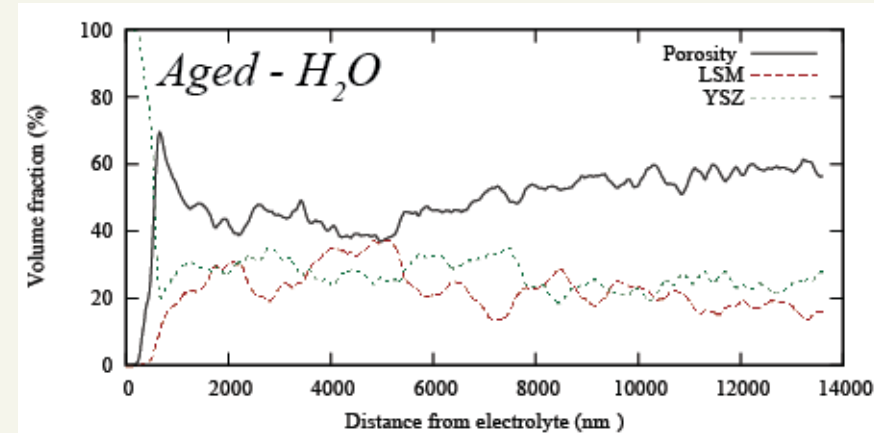
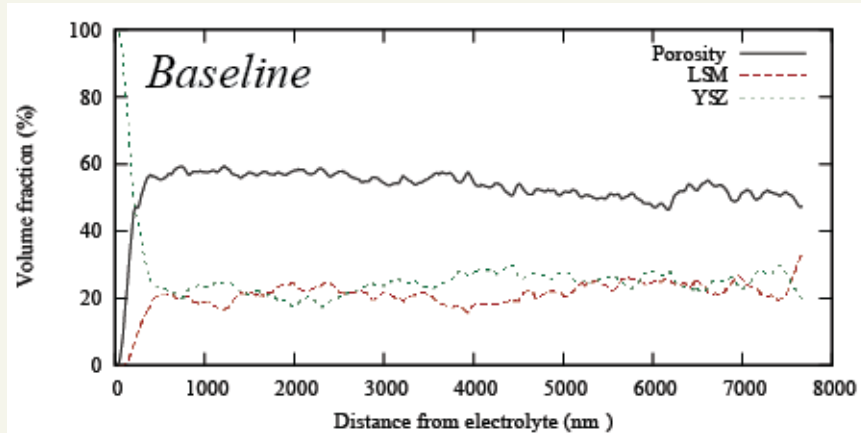


- Phase solid fractions remain similar to expected values (from source materials)

- Porosity is decreased in both aged samples
 - Additional sintering occurring during aging process

	Exp. YSZ	Exp. LSM	Obs. YSZ	Obs. LSM
Baseline			0.527	0.473
Aged-dry	0.52	0.48	0.503	0.497
Aged-H ₂ O			0.542	0.458

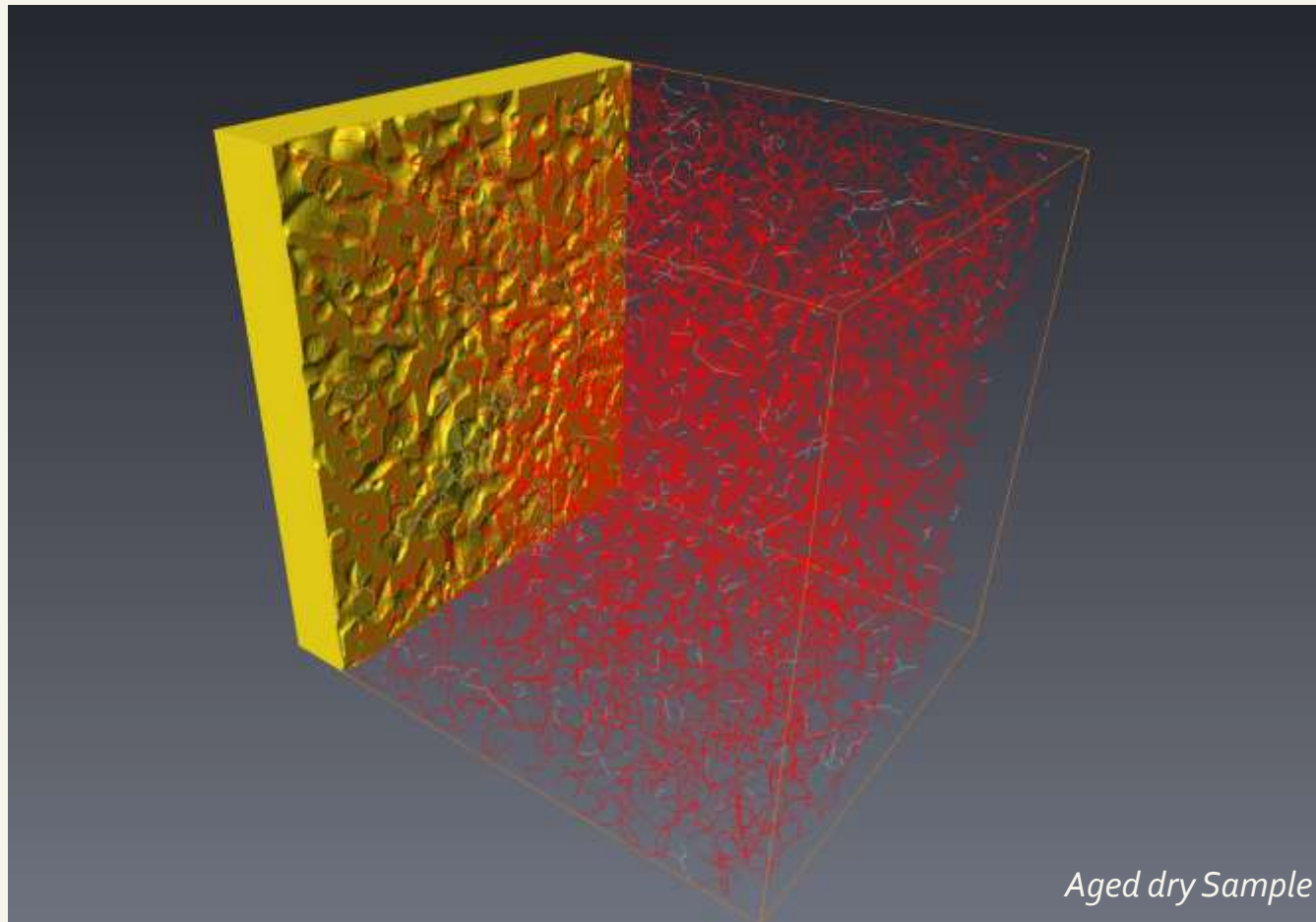
Results – Graded phase fractions



- Plotting phase volume fraction as a function of distance from cathode/electrolyte interface
 - Influence of aging on phase distribution
- Greater variability in aged samples:

σ of phase fraction	Pore	LSM	YSZ
Unaged	2.51	2.24	2.83
Aged-dry	3.75	5.45	2.61
Aged-H ₂ O	6.73	5.92	3.92

Results – Phase connectivity

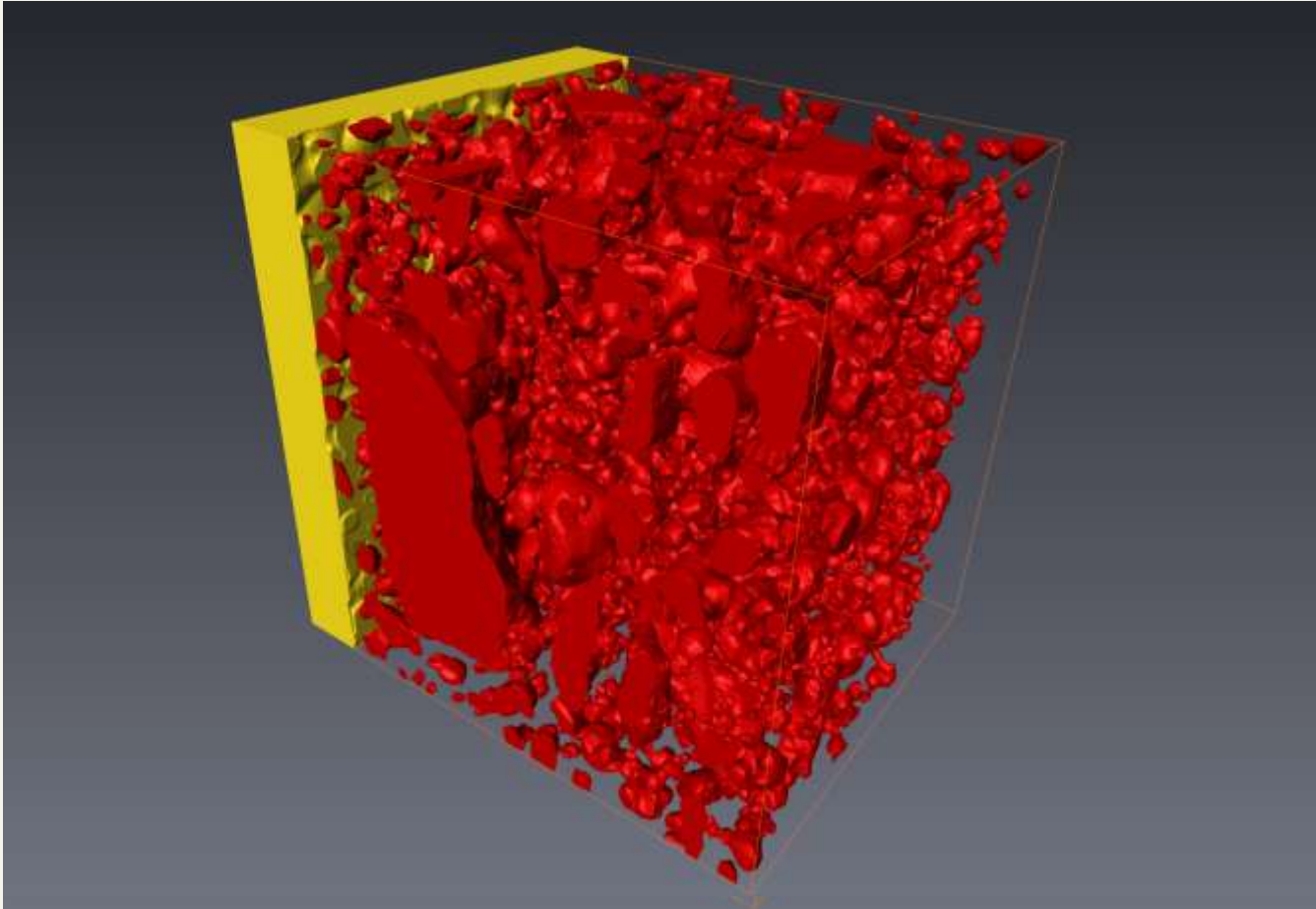


Skeletonization of individual phases reveals connectivity

Isolated LSM networks within volume

Contiguous YSZ and pore networks

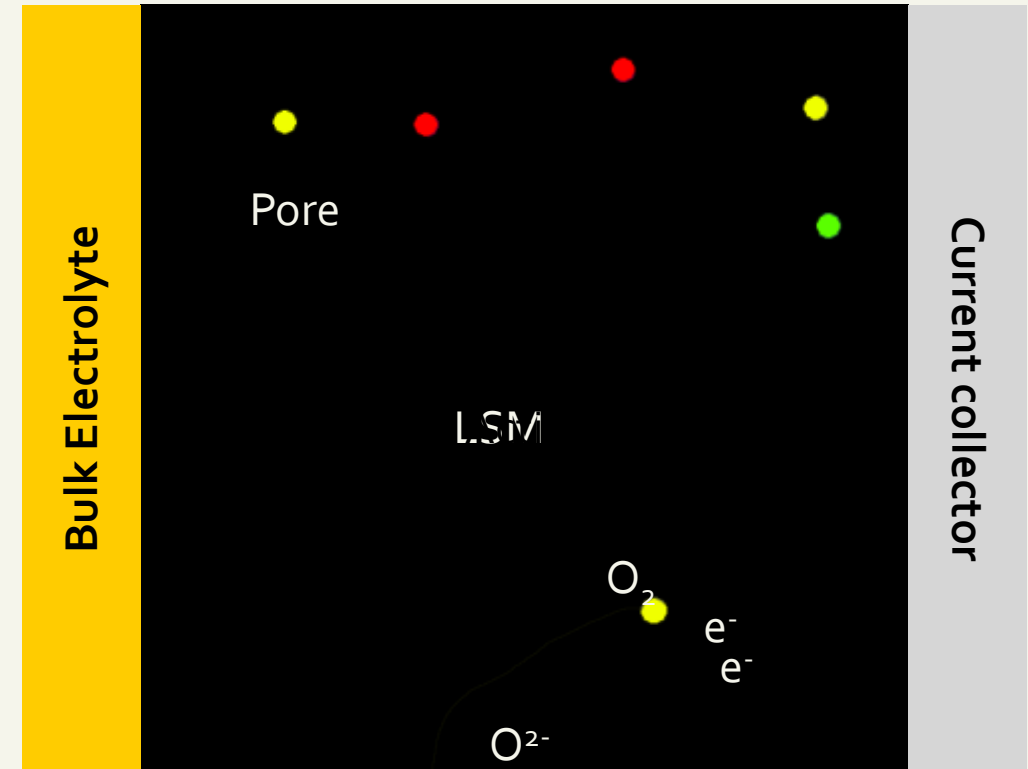
Results – Phase connectivity



- How to quantify the connectivity?
 - Average degree of connectivity ($\langle k \rangle$)
 - Max degree of connectivity
 - Percent of length connected to contiguous pathway
- All of these depend on how the skeleton is calculated
 - Two main options within Avizo:
 - Auto skeleton module
 - Centerline tree module
 - Is either really appropriate?

Triple phase boundary (L_{TPB}) determination

- Intersection of three phases is necessary for the oxygen reduction reaction to occur:
 - ORR: $\frac{1}{2}O_2 + 2e^- \leftrightarrow O^{2-}$
 - This quantity can be directly related to cell performance
- Within analysis volume, a phase and boundary site can be described as **active**, **inactive**, or **unknown**
- Labels depend on connection to edges
 - Unknown have at least 1 border with edges (dead-end)
 - Active have two borders across a dimension (transverse)
 - Inactive networks have no intersection with an edge (isolated)
- Collaboration with Scientific Applications and Visualization Group at NIST



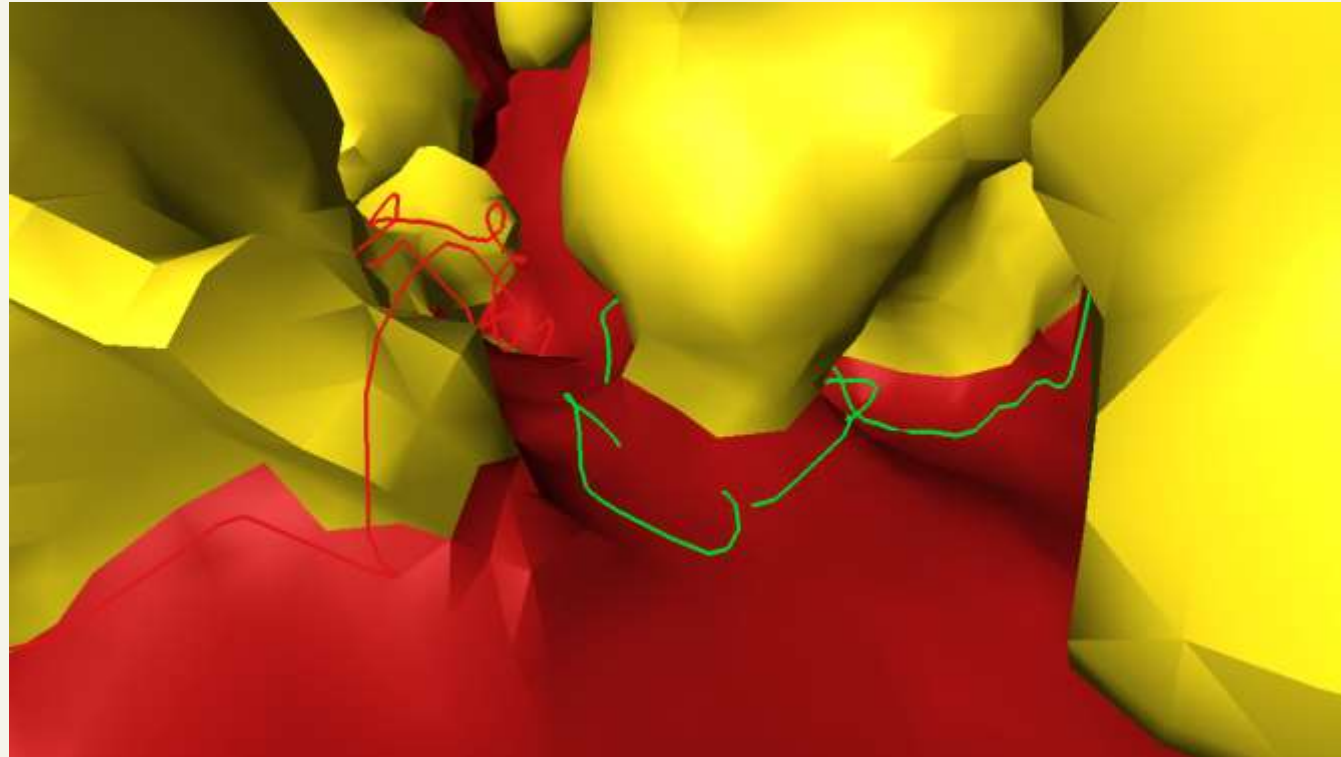
Results – Triple phase boundaries



To be active, all three connected components of the TPB must be contiguous throughout the volume

(which requires a large enough sampling volume to be representative)

Results – Triple phase boundaries

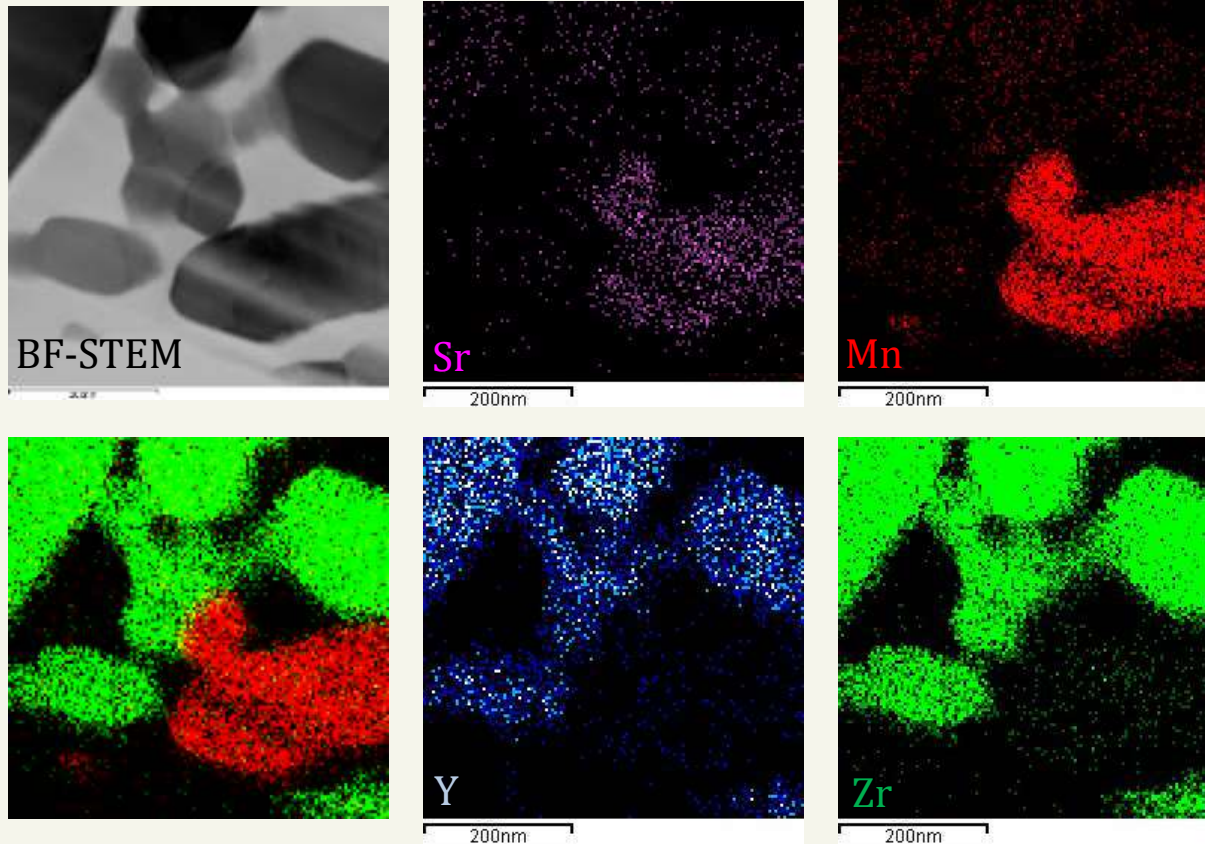


Results – Triple phase boundaries

<i>*Relative to baseline</i>	Baseline	Aged-dry	H₂O
Active TPB [norm]	100%	32.5% *	27.5% *
Total ρ_{TPB} [$\mu\text{m}/\mu\text{m}^3$]	19.2	9.69	8.57

- Absolute quantifications are forthcoming, but we can analyze differences between the two samples, which show decrease in all L_{TPB} values upon exposure to H₂O contamination

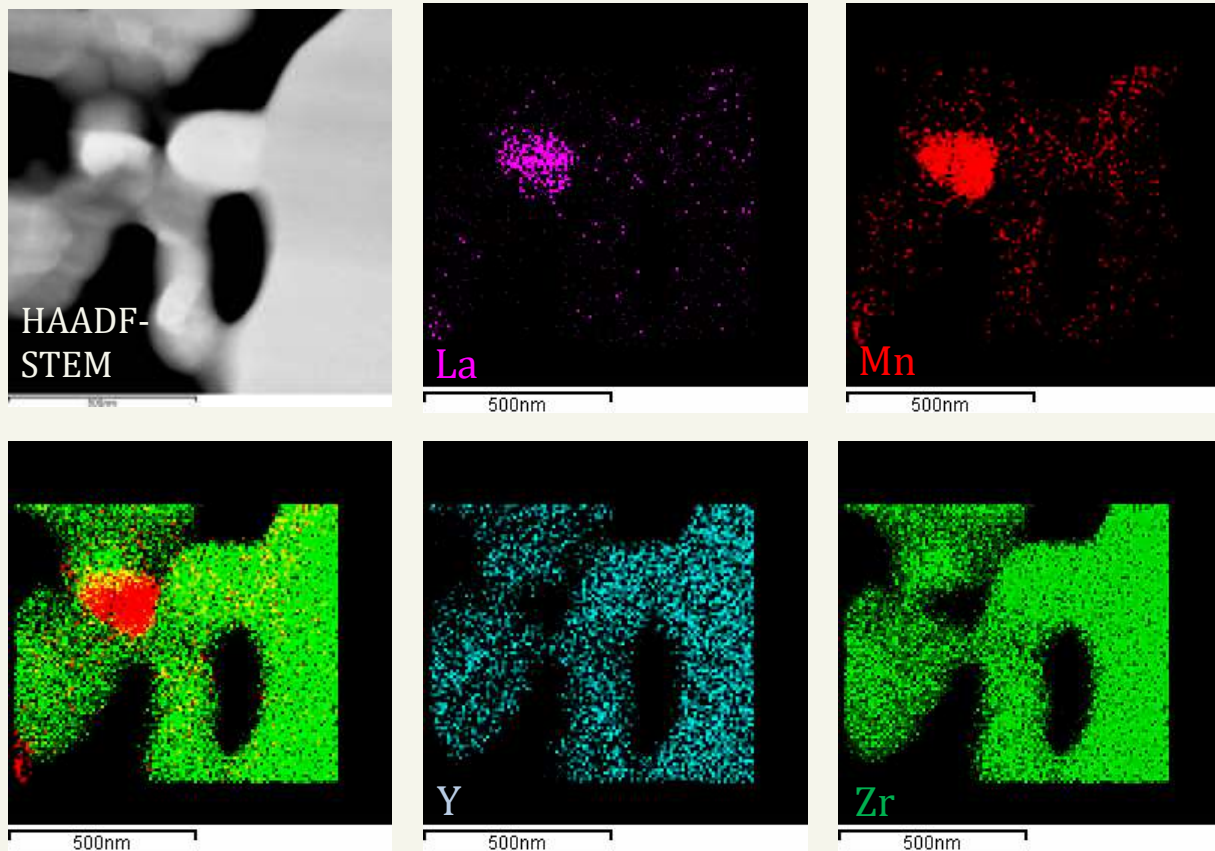
Results – Preliminary TEM/EDS work



STEM-EDS maps of
Baseline SOFC
cathode near
electrolyte interface

- Distinct particles of LSM and YSZ

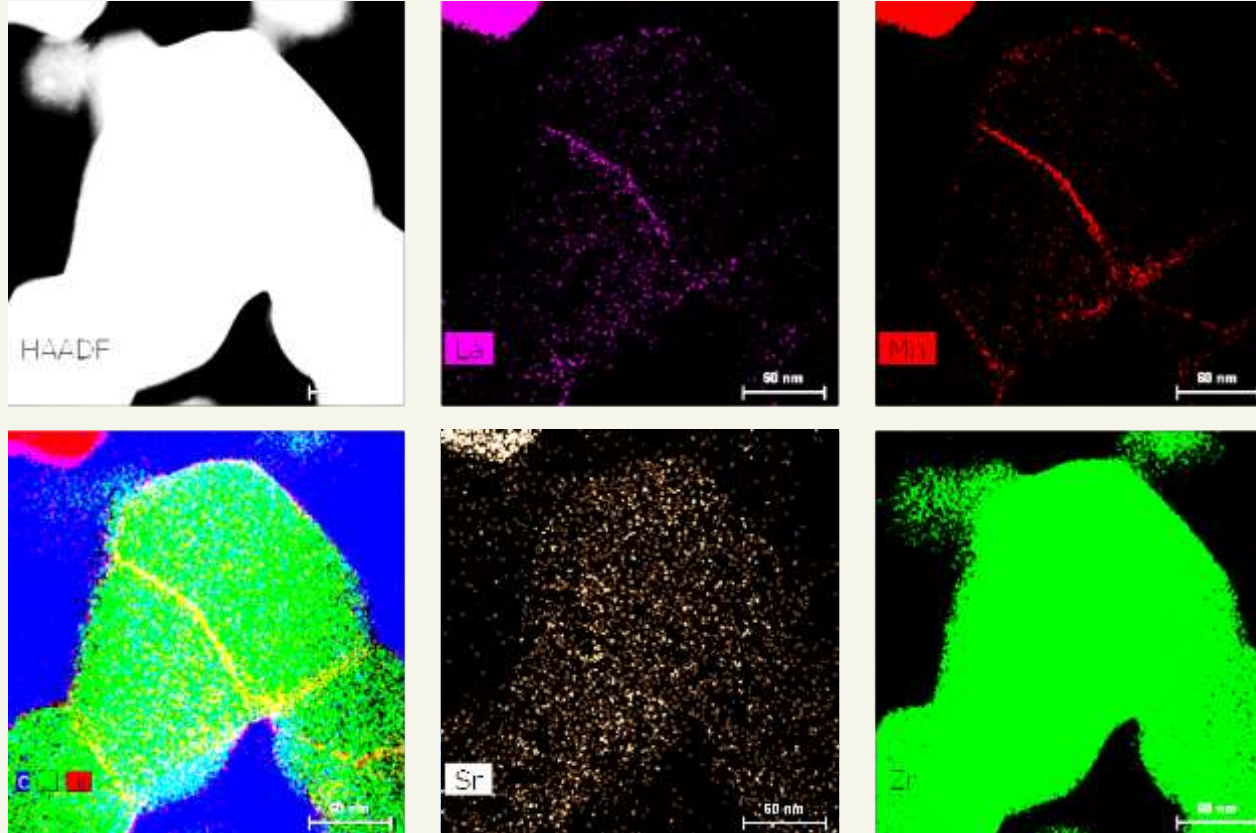
Results – Preliminary TEM/EDS work



STEM-EDS maps of
Aged-dry SOFC
cathode near
electrolyte interface

- Still distinct particles of LSM and YSZ
- Perhaps more Mn distributed throughout YSZ

Results – Preliminary TEM/EDS work



Collected on FEITitan™ G2 80-200 ChemiSTEM, courtesy of FEI Company

STEM-EDS maps
Aged-H₂O SOFC
cathode

- Distinct particles of LSM and YSZ
- Segregation of La and Mn at YSZ grain boundaries
- Sr is not localized at boundaries

Summary

Conclusions

- We have developed and refined methods using both Avizo Fire and external calculations to quantify 3D microstructure of solid oxide fuel cell cathodes
- At the conditions tested, additional sintering causes changes in phase fractions throughout cathode
- $\rho_{\text{TPB,active}}$ decreases when aged under H_2O contamination
- Segregation of La and Mn to YSZ grain boundaries in aged- H_2O (but not Sr)

Upcoming Work

- Observe sample aged in higher H_2O concentrations to confirm effects
- Analyze and quantify composition of segregation products using TEM/EELS
- Correlate with corresponding EIS data

Acknowledgements



NSF GRFP, Grant No. DGE 1322106
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NIST



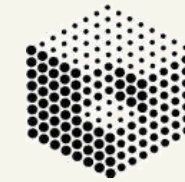
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THANK YOU

Questions and comments?